ESTIMATIONS ON  $\mathbf{J}_{\mathbf{IC}}$  IN VIEW OF FRACTOGRAPHIC ANALYSIS

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The exact location of the initiation point in the stable crack propagation conditions at the  $\rm J_{IC}$  test was investigated. A modification of the ASTM test procedure was proposed; it is based on the philosophy that the critical stretch zone width (SZW) is a very representative parameter reflecting direct indications of the ductile tearing onset. The results of the modified method were compared with those of the ASTM procedure. The stretch zone width approach has been shown to be a suitable back-up procedure usefully applicable for the verification of the standard  $\rm J_{IC}$  values.

## INTRODUCTION

It has been reported (1) that the blunting line obtained according to the ASTM E 813 procedure is not fully representative in the case of  $\rm J_{IC}$  determination of heat treated steels since the  $\rm J_{IC}$  values obtained are usually greater than those determined by other methods (e.g. JSME S 001-81). In this work, the ASTM E 813 test commonly with the modified procedure based on the average critical stretch zone width  $\Delta a_{\rm SZ}$  fractographic measurements as applied for the  $\rm J_{IC}$ 

## RESULTS AND DISCUSSION

The low alloy wrought steel ČSN 41 6341 of a composition equivalent to that of AISI 4340 was heat treated by austenitizing at 840°C for 0,5 h, oil quenched and tempered at temperatures in the range 500-700°C with corresponding  $\sigma_{\rm y}$  values from 1220 down to 747 MPa.

Three-point bend specimens with L-T orientation

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were prepared according to the ASTM standard E 813. Multiple specimen  $J_R$ -curve methods were used to determine  $J_{IC}$  fracture toughness value: 1) ASTM E 813 method (Fig.1(a)), and 2) Modified E 813 method (Fig.1(b)). In this method, required conditions for qualifying R-curve regression line are exactly the same as those in the E 813. The initiation point of slow-stable crack growth differs, however, from E 813 and it is defined by the intersection point between a linear regression line and the average critical SZW line (Fig.1(b)). It is evident from TABLE 1

TABLE 1

Tempering temp./OC/	$\rm J_{IC}$ /kJm $^{-2}$ /		∆ā <sub>sz</sub>	dJ da
	Method 1	Method 2	//um/	da /MPa/
500 600 650 700	39 68 86 124	39 67 82 117	15.6 24.0 25.5 41.2	83 130 197 210

that the crack growth initiation evaluated according to the modified method (Fig.1(b)) occurs at a J-value negligibly smaller than the JIC value determined by the ASTM E 813. For the AISI 4340 steel Putatunda and Rigsbee (2) earlier reported a much greater  $\Delta a_{\rm SZ}$  value than that presented above. The differences in various observed critical SZW values could have their origin (likely) mainly in the discrepancy in fractographic interpretation of the SZW. Most probably, in the given case as the reported SZW both the blunted and the fine dimpled shear zones (along logarithmic spirals) were taken. This view is supported by the results of the extensive round robin test presented in (3).

The  $J_{\rm IC}$  value of the material studied was also determined by the method 3 (Fig.1(c)). There are several alternatives for constructing the blunting line; shown in Fig.1(c). In our work, these different approaches were compared and assesed. Altough the results obtained in this way were not in full accord with those, given in Table 1, we believe that the proposed method can be effectively used for the failure analysis and/ /or as a single specimen  $\mathtt{J}_{\mathrm{IC}}$  determination.

## REFERENCES

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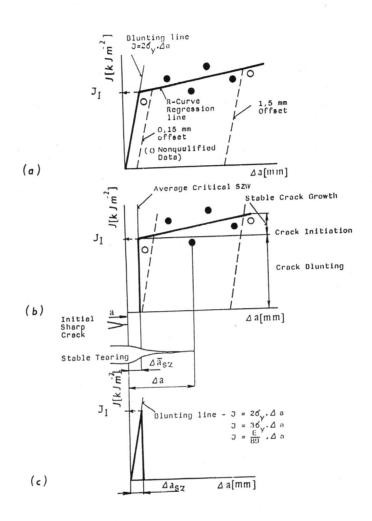


Fig. 1 Test procedure for J<sub>TC</sub>: (a) ASTM R-curve method,(b) Modified ASTM method,(c) SZW method.